# **RSC Advances**

## PAPER

Check for updates

Cite this: RSC Adv., 2017, 7, 43227

Received 28th July 2017 Accepted 30th August 2017 DOI: 10.1039/c7ra08334c

rsc.li/rsc-advances

### Introduction

Owing to the depletion of fossil fuels and environmental issues, renewable energy technologies such as solar, wind, biomass, and tidal power have attracted much attention globally.<sup>1,2</sup> The clean energy from these renewable resources can be naturally replenished on the human timescale. However, power generation by such systems is uncontrolled and may be operative at a time when no additional power is required. To achieve a stable, on-demand energy supply, large-scale electrical energy storage (EES) systems are indispensable.3 The design of redox flow batteries (RFBs) can be flexibly tailored for target applications, and RFBs offer long-term electrochemical stability given that the active redox-couples with no host structure are dissolved in two separate electrolytes that are individually stored in containers of the desired size, making RFBs promising candidates as large-scale EESs.4-6 The operating voltage of RFBs is tunable depending on the redox couples used, some of which are operational within the water splitting voltage window (~1.23 V) owing to their large overpotential.<sup>7,8</sup> However, when

Min Eui Lee,<sup>a</sup> Hyoung-Joon Jin<sup>b</sup> \*<sup>a</sup> and Young Soo Yun\*<sup>b</sup>

All-vanadium redox flow batteries (VRFBs) have attracted a great deal of attention as large-scale energy storage devices. However, VRFBs suffer from unfavorable energy efficiency, originating from large polarization effects that influence the redox reaction of the vanadium ions. Although the catalytic effects of the oxygen and nitrogen functional groups in carbon-based electrodes on the redox reactions of the vanadium ion have been documented in several reports, the specific catalytic effects are not understood due to the lack of systematic studies. In this study, pyroproteins having dual functionality (i.e., nitrogen and oxygen functional groups) were coated onto the surface of conventional carbon felts (the composite is hereafter termed P-CFs) and their catalytic effects on the  $V^{2+}/V^{3+}$  redox reaction were characterized and compared with those of carbon felts possessing a coating layer with only oxygen or nitrogen functional groups. The synergistic catalytic effects of nitrogen and oxygen in the pyridonic structure of the P-CFs led to the lowest anodic and cathodic peak potential separation of ~0.17 V among all the evaluated samples, having various chemical configurations with oxygen or nitrogen only. This study provides insight for the design of carbon-based active electrode materials in VRFBs and their feasibility was demonstrated in VRFB full-cells.

> different types of redox couples are utilized as the two separated electrolytes, self-discharge originating from ion-crossover leads to decay of the energy efficiency with successive cycles.9,10 Therefore, all-vanadium redox flow batteries (VRFBs) that employ vanadium ions in both the anolyte and catholyte were proposed by Skyllas-Kazacos and coworkers, who achieved considerably reduced ion-crossover.<sup>11,12</sup> In addition, the redox reactions between  $V^{2+}/V^{3+}$  (-0.26 V  $\nu s$ . SHE) and  $VO^{2+}/VO_{2}^{-1}$ (1.0 V vs. SHE) have a high redox potential ( $\sim$ 1.26 V), leading to high energy density.13,14 Despite the merits of VRFBs, the decline of the cell efficiency due to intrinsic polarization is regarded as a major limitation of VRFBs, hindering their extensive application.15,16 The cell polarization of VRFBs mainly originates from the charge-transfer resistance between the vanadium ions and the electrode, wherein the V(II)/V(III) and V(IV)/V(V) redox reactions are highly affected by the surface properties of the electrodes.17-21 Carbon felt (CF) has typically been used as an electrode for VRFBs because of its three-dimensional macroporous structure that facilitates mass transport, high electrical conductivity, and chemical stability in strongly acidic electrolytes; CF is also amenable to mass production.<sup>21-24</sup> However, owing to the poor electrochemical activity and kinetic reversibility, several studies have focused on alleviating activation polarization of CF by surface modification with species such as redox-active heteroatoms and/or nanocomposites with metalbased electrochemical catalysts.15,16,20,21,23-26 Although metalbased catalysts such as Pt,<sup>27</sup> Mn<sub>3</sub>O<sub>4</sub>,<sup>28</sup> WO<sub>3</sub>,<sup>29</sup> CeO<sub>2</sub>,<sup>30</sup> and ZrO<sub>2</sub>



View Article Online

View Journal | View Issue

<sup>&</sup>lt;sup>a</sup>WCSL (World Class Smart Lab) of Green Battery Lab., Department of Polymer Science and Engineering, Inha University, Incheon 22212, South Korea. E-mail: hjjin@inha.ac.kr <sup>b</sup>Department of Chemical Engineering, Kangwon National University, Samcheok 25913, South Korea. E-mail: ysyun@kangwon.ac.kr

<sup>†</sup> Electronic supplementary information (ESI) available: Tables S1 to S8 and Fig. S1 to S10. See DOI: 10.1039/c7ra08334c

Synergistic catalytic effects of oxygen and nitrogen functional groups on active carbon electrodes for all-vanadium redox flow batteries†

(ref. 31) exhibit high redox-activities with vanadium redox couples, these species exhibit poor stability under acidic conditions and/or require a relatively complex process with expensive precursors, which is undesirable for large-scale EESs. On the other hand, the introduction of redox-sensitive functional groups onto the surface of CF through facile surface modification using well-developed chemistry leads to significant mitigation of the overpotential originating from the major activation polarization process.<sup>20,21,23,25,32-36</sup> Kim et al. reported that CF with oxygen functional groups (e.g., hydroxyl or carboxylic acid groups) introduced onto the surface via combined corona discharge and hydrogen peroxide was suitable as a redox host for the VO<sup>2+</sup>/VO<sub>2</sub><sup>+</sup> couples.<sup>35</sup> It was also reported that the redox activity could be improved by co-doping nitrogen atoms and oxygen groups onto the surface of CF.19,36 Despite these results and though the effects of nitrogen, oxygen, and their hybrid groups on carbon-based electrodes have been demonstrated in several reports, systematic studies of their catalytic effects remain insufficient.17,19,20,23,26,32,33,36-38 Therefore, it is difficult to understand which types of heteroatoms or chemical configurations are most effective, what is the relationship between the heteroatom content and the catalytic effect, and whether there is a synergistic effect between the oxygen and nitrogen functional groups. Furthermore, it is noteworthy that the catalytic reactions of the redox-active heteroatoms with V2+/V3+ and VO2+/VO2+ progress independently, leading to different catalytic effects on both sides.<sup>23</sup> In other words, separate surface modification is needed for the anode and the cathode in order to enhance the catalytic effects. Nevertheless, most of the literature has focused on development of the cathode for the VO<sup>2+</sup>/VO<sub>2</sub><sup>+</sup> redox couples, whereas there are few reports concerning the anode.<sup>15,39,40</sup> Considering that the theoretical redox voltage of the  $V^{2+}/V^{3+}$  couples is 0.26 V lower than that of the hydrogen evolution reaction (0 V vs. SHE) in acidic aqueous electrolyte, competition between the desired reaction and water splitting is inevitable; this may have a negative impact on the reaction kinetics and efficiency.<sup>39</sup>

In this study, thin pyroprotein coating layers containing numerous oxygen and nitrogen heteroatoms are introduced onto the surface of CF (nitrogen and oxygen co-doped sample; P-CFs), and their catalytic effects on the redox reaction of  $V^{2+}/V^{3+}$  couples for VRFBs are investigated. Further, to confirm the inter-relation between the catalytic effects of the surface species, oxygen or nitrogen-doped carbonaceous coating layers are prepared on the surface of CF via glucose (oxygen-doped sample; O-CFs) or melamine (nitrogen-doped sample, N-CFs) adsorption with subsequent thermal treatment. It is demonstrated that the pyroprotein coated CFs exhibits a significantly lower anodic and cathodic peak potential separation ( $\Delta E_p$ ) of  $\sim$ 0.17 V, compared with those of O-CFs and N-CFs ( $\sim$ 0.32 and  $\sim$ 0.24 V, respectively); even with the P-CF sample having a lower heteroatom content than those of the O-CFs and N-CFs, a superior  $\Delta E_{\rm p}$  value is achieved. These results clearly demonstrate the synergistic catalytic effects of the oxygen and nitrogen heteroatoms on the  $V^{2+}/V^{3+}$  redox reactions. Furthermore, fullcell VRFBs based on P-CFs exhibit enhanced energy efficiency,

rate capability, and stable cycling behavior compared with those based on CFs.

### **Experimental section**

#### Preparation of regenerated silk fibroin solution

Cocoons from *B. mori* silkworms were obtained from the Uljin Farm, South Korea. To extract the glue-like sericin protein and impurities, the silkworm cocoons were boiled for 25 min in an aqueous solution of 0.02 M Na<sub>2</sub>CO<sub>3</sub> (*i.e.*, degumming process). After washing with deionized (DI) water several times, the resulting silk fibroin was dried at room temperature for 3 days. The regenerated silk protein solutions were prepared as described elsewhere.<sup>41</sup> Briefly, the degummed silk fibroin was dissolved in an aqueous solution of 9.3 M LiBr (99%; Sigma-Aldrich, St. Louis, MO, USA) at 60 °C for 6 h. The silk protein solution was dialyzed in water using a Slide-a-Lyzer dialysis cassette (MWCO 3500; Pierce) for 2 days. The final concentration of the regenerated silk fibroin (RSF) in the aqueous solution was approximately 8 wt%.

#### Preparation of P-CFs, O-CFs, and N-CFs

P-CFs, O-CFs, and N-CFs were prepared by simple solutioncoating of pristine CFs (XF-30A, thickness: 4.3 mm, Toyobo, Japan), followed by thermal treatment at the desired temperature. The CF was dip-coated in an aqueous 0.5 wt% solution of RSF, glucose (99.5%, Sigma-Aldrich), or melamine (99%, Sigma-Aldrich) and dried in a vacuum oven at 30 °C for 3 h. The RSF, glucose, or melamine-coated CFs were heated at different temperatures (400, 600, 800, 1000, and 1200 °C) in a tube furnace. A heating rate of 5 °C min<sup>-1</sup> and an Ar flow rate of 200 mL min<sup>-1</sup> were applied; the samples were held at the final temperature for 2 h.

#### Characterization

The morphologies of the samples were observed using fieldemission scanning electron microscopy (FE-SEM, S-4300SE, Hitachi, Japan). X-ray diffraction (XRD, Rigaku DMAX-2500) analysis of the samples was performed using Cu-K $\alpha$  radiation (wavelength  $\lambda = 0.154$  nm) at 40 kV and 100 mA. Raman spectra were acquired using a continuous-wave, linearly polarized laser (514.4 nm wavelength, 2.41 eV, 16 mW power). The laser beam was focused by a 100× objective lens, resulting in a spot with a diameter of ~1 µm. The acquisition time and number of circulations used to collect each spectrum were 10 s and 5, respectively. The chemical compositions of the samples were examined by XPS (AXIS-HIS, Kratos Analytical, Japan) using monochromated Mg-K $\alpha$  radiation ( $h\nu = 1486.6$  eV).

#### **Electrochemical characterization**

The electrochemical performance of the samples and their flowcell devices were characterized with the aid of a potentiostat (PGSTAT302N, Autolab), Wonatech automatic battery cycler, and single-cell test equipment (Standard Cell – Blue, Standard Energy Co., Ltd., Republic of Korea). For the half-cell experiments, a hand-made, three-electrode configuration system was

#### Paper

used,<sup>42</sup> in which P-CFs, O-CFs, or N-CFs with a diameter of 5 mm (area: 0.196 cm<sup>2</sup>) excess CF, and Ag/AgCl were used as the working, counter, and reference electrodes, respectively. VOSO<sub>4</sub> (0.1 M, 99.5%, Sigma-Aldrich) was dissolved in 2 M H<sub>2</sub>SO<sub>4</sub> (96.5%, Sigma-Aldrich) and used as the electrolyte for the half-cell tests. For the full-cell experiments, a single cell was assembled with copper current collectors, bipolar plates, and P-CF//CF or CF//CF electrode pair, and cation exchange membranes (Nafion 115, DuPont Co., Ltd., USA). As a pretreatment, the membrane was immersed in DI water for 24 h. The working electrodes with an area of 9 cm<sup>2</sup> were used as both the cathode and anode. The electrolyte was prepared by dissolving 1.6 M VOSO<sub>4</sub> in 4 M H<sub>2</sub>SO<sub>4</sub> solution and the electrolyte volume in each of the catholyte and anolyte tanks was 60 mL.

### **Results and discussion**

The morphologies of the CFs and P-CFs-400 samples (where the heat-treatment temperature is indicated at the end of the nomenclature, e.g. -400, -600, etc.) are shown in Fig. 1(a) and (b). Both samples have a similar macroporous structure and smooth topologies, suggesting homogeneous coating of the pyroprotein layer on the surface of the CFs. It was also confirmed that N-CFs-400 and O-CFs-400 exhibited no morphological differences compared with CFs and P-CFs (Fig. S1<sup>†</sup>). The surface carbon microstructure of the samples was characterized by acquiring the Raman spectra, which showed characteristic D and G bands of hexagonal carbon structure centered at  $\sim$ 1350 and  $\sim$ 1580 cm<sup>-1</sup>, respectively. The D and G bands originate from the intrinsic phonon mode of the infinite aromatic ring with A1g symmetry that is activated by the structural disorder and the hexagonal carbon structure related to the E<sub>2g</sub> vibration mode of the sp<sup>2</sup>-hybridized C atoms, respectively. Hence, the D to G intensity ratio  $(I_D/I_G)$  indicates the approximate size of the poly-hexagonal carbon structures. The  $I_D/I_G$  ratios of P-CFs-400, N-CFs-400, and O-CFs-400 were  ${\sim}1.02,\,{\sim}1.00,$  and  ${\sim}1.01,$  which are higher than that (0.86) of CFs, indicating that the surface carbon structures of the P-CFs-400, N-CFs-400, and O-CFs-400 samples have more defects. In addition, the G bands of CFs were distinctively shifted to higher frequencies (blue-shift) in the case of P-CFs-400, N-CFs-400, and O-CFs-400. The blue-shift of the G band suggests that the Raman-inactive non-zero phonon density of states above the G band become active owing to phonon confinement caused by the created defects.<sup>43</sup>

These results also confirm that the properties of the surface carbon in the samples were altered by introducing the coating layers onto the surface of the CFs. In contrast, the X-ray diffraction (XRD) patterns of the samples showed similar broad peaks at  $25.3^{\circ}$ , corresponding to the graphitic (002) structure, suggesting that the bulk carbon structures remained unchanged. The modified surface properties of P-CFs-400, N-CFs-400, and O-CFs-400 were characterized using X-ray photoelectron spectroscopy (XPS), as shown in Fig. 2. The O 1s spectra of P-CFs-400 and O-CFs-400 show similar peaks corresponding to chemical structures with C-O and C=O bonds (Fig. 2(a)), although the quantity of oxygen and the bonding ratios were different for the samples (Fig. 2(b) and (c)). The C/O ratio of O-CFs-400 was 4.5, which is much smaller than those of CF (15.8) and P-CF (8.6). The oxygen content of O-CFs and P-CFs gradually decreased with an increase in the heat-treatment temperature (HTT) from 400 to 1200 °C (see Fig. S2 and S3<sup>†</sup>). The N 1s spectra of P-CFs-400 and N-CFs-400 show similar peaks of pyridinic-N and N-oxide structures at  $\sim$ 398.3 and  $\sim$ 403.2 eV, respectively (Fig. 2(d)). Further, the major chemical bonds gave rise to peaks at  $\sim$ 400.6 and  $\sim$ 400.1 eV for P-CFs-400 and N-CFs-400, respectively. It is noteworthy that although the main peaks of P-CFs-400 and N-CFs-400 were located in similar positions, the peaks indicate different chemical structures. Due to the presence of edge sites of poly-hexagonal carbon structures with numerous oxygen and nitrogen, the main peak of P-CFs-400 could originate from the pyridonic-N structure,

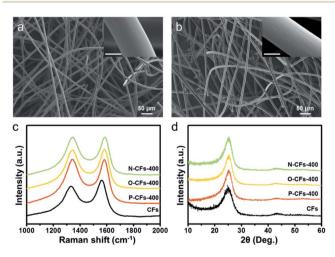
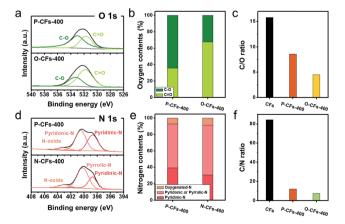


Fig. 1 FE-SEM images of (a) CFs and (b) P-CFs-400 (inset: FE-SEM images, scale bar: 5  $\mu$ m). (c) Raman spectra and (d) XRD patterns of CFs, P-CFs-400, O-CFs-400, and N-CFs-400.



**Fig. 2** XPS analysis for P-CFs-400, O-CFs-400, and N-CFs-400. (a) High-resolution O 1s spectra, (b) oxygen chemical composition and (c) content of oxygen functional groups for P-CFs-400 and O-CFs-400. (d) High-resolution N 1s spectra, (e) nitrogen chemical composition and (f) content of nitrogen functional groups for P-CFs-400 and N-CFs-400.

whereas for N-CFs-400 containing numerous nitrogen species and negligible oxygen, the major peak corresponds to the pyrrolic-N structure. Except the chemical configuration of the major peak, there was no distinct difference in the nitrogen bonding of P-CFs-400 and N-CFs-400 (Fig. 2(e)). As shown in Fig. 2(f), the CFs sample had a negligible nitrogen content corresponding to a C/N ratio of 84.5, whereas the P-CFs-400 sample had a considerably high nitrogen content (C/N ratio of 12.1), which is about 7 times higher than that of CFs (Fig. 2(f)). Moreover, the nitrogen content of N-CFs-400 is around 1.5 times higher than that of P-CFs-400, corresponding to a C/N ratio of 7.5. The nitrogen content of P-CFs and N-CFs also decreased with a gradual increase in the HTT from 400 to 1200 °C (see Fig. S3<sup>†</sup>). The specific C/O and C/N ratios of the samples are summarized in Tables S1-S3.† Additionally, the modified surfaces of P-CFs-400, N-CFs-400, and O-CFs-400 exhibited good wettability, which is in clear contrast with that of CFs (Fig. S4<sup>†</sup>).

The electrochemical catalytic activity of CFs, P-CFs-400, O-CFs-400, and N-CFs-400 was evaluated vs. Ag/AgCl in the voltage window of 0 to -1.0 V by using a hand-made threeelectrode configuration system with 0.1 M VOSO<sub>4</sub> dissolved in 2 M  $H_2SO_4$  solution as the electrolyte (Fig. S5<sup>†</sup>). The cyclic voltammograms (CVs) of the samples characterized at 5 mV  $s^{-1}$ present distinct redox peaks with different  $\Delta E_{p}$  values and peak current densities (Fig. 3(a)). P-CFs-400 gave rise to the lowest  $\Delta E_{\rm p}$  value (~170.9 mV) of all the characterized samples, including N-CFs-400 and O-CFs-400 with  $\Delta E_{\rm p}$  values of 238.0 and 352.4 mV, respectively. These results reveal that the catalytic effects of active carbon materials on the redox reaction of  $V^{2+}/V^{3+}$  couples are not just dependent on the quantitative

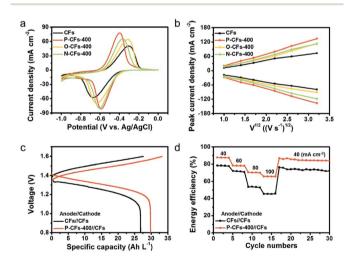


Fig. 3 Electrochemical performance characterized using half-cell and full-cell configurations. (a) CV curves at a sweep rate of 5 mV  $s^{-1}$  and (b) peak current densities at different sweep rates from 1 to 10 mV  $\rm s^{-1}$ for CFs, P-CFs-400, O-CFs-400, and N-CFs-400 over a voltage window of -1.0 to 0 V vs. Ag/AgCl in an electrolyte of 0.1 M VOSO4 dissolved in 2 M H<sub>2</sub>SO<sub>4</sub>; half-cell was used. VRFB single-cell operation based on CFs//CFs and P-CFs-400//CFs: (c) galvanostatic charge/ discharge profiles at a current density of 60 mA  $cm^{-2}$  and (d) energy efficiency for continuous charging-discharging cycles at different current densities over a voltage window of 0.8-1.6 V in an electrolyte of 1.6 M VOSO<sub>4</sub> dissolved in 4 M H<sub>2</sub>SO<sub>4</sub>.

values of the active chemical species because O-CFs-400, having the highest heteroatom content, gave rise to a relatively poor  $\Delta E_{\rm p}$  value. In the specific case of O-CFs-400, although the oxygen content declined significantly with increasing HTT, the  $\Delta E_{\rm p}$  values obtained at the same sweep rate of 5 mV s<sup>-1</sup> showed a moderate increase from 323.4 to 359.1 mV, corresponding to 11% (Table S4<sup>†</sup>). The overall  $\Delta E_p$  values of O-CFs were similar to that (363.0 mV) of CFs (Table S4<sup>†</sup>). In contrast, the  $\Delta E_p$  values of N-CF declined continuously from 238.0 to 352.4 mV as the C/N ratio increased, corresponding to an  $\sim$ 82% decrease (Table S5<sup>†</sup>). Therefore, from a qualitative point of view, the superiority of nitrogen groups compared with oxygen groups can be verified from these results. In addition, the  $\Delta E_{\rm p}$  value of P-CFs is much smaller than them of previously reported results (Table S7<sup>†</sup>). However, in order to account for the highest performance of P-CFs-400 with a relatively low nitrogen content compared with N-CFs, a new interpretation of the catalytic effect on the  $V^{2+}/V^{3+}$ redox reaction is required. Considering the chemical structures of N-CFs and P-CFs, the better electrochemical performance of P-CFs may be induced by the pyridonic structure of P-CFs-400. In particular, a clear difference was observed in the ex situ O 1s spectra of P-CFs-400 versus O-CFs-400 during the redox reaction, as shown in Fig. S6.<sup>†</sup> In the XPS O 1s spectra of P-CFs-400 acquired after the reduction reaction at -0.8 V, a strong shift (of about 1.1 eV) of the O 1s peak toward lower binding energy was observed, whereas the XPS O 1s spectra of O-CFs acquired under the same conditions showed a weak shift (about 0.3 eV) (Fig. S7<sup>†</sup>). This result suggests a strong increase in the electron density of the oxygen groups of P-CFs-400, which could be accompanied by the redox reaction of the  $V^{2+}/V^{3+}$ couple. Although the oxygen chemical configurations of P-CFs-400 and O-CFs-400 are similar, the peaks of the oxygen groups of P-CFs-400 were more strongly shifted, which is not explained by a redox reaction involving just oxygen groups. Although it remains inconclusive, as shown in the schematic illustration of Fig. 4, the nitrogen beside the oxygen on the edge site of the poly-hexagonal carbon structures possibly can work in tandem

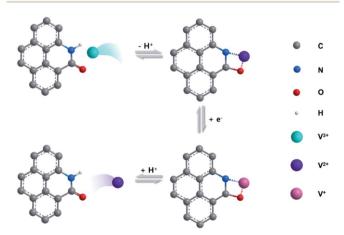


Fig. 4 Schematic image exhibiting possible mechanism of redox reaction of  $V^{2+}/V^{3+}$  redox couple on pyridonic structure of P-CFs. Oxygen and nitrogen functional groups can provide a catalytic effect for the redox reaction by their synergistic behavior.

(cc) BY-NC

#### Paper

with the oxygen in the redox reaction of  $V^{2^+}/V^{3^+}$  couples. The synergistic catalytic effects of nitrogen and oxygen have been observed in alkali cation redox reactions.<sup>44,45</sup>

Furthermore, the slopes of the square root of the sweep rate versus oxidation and reduction peak current density plots have larger value for P-CFs-400 than for O-CFs-400, N-CFs-400, and CFs, indicating better mass transfer in P-CFs-400 (Fig. 3(b)). Moreover, the catalytic effects of the samples on the  $VO^{2^+}/VO_2^{-1}$ couple are completely different from their catalytic effects on the redox reaction of the  $V^{2+}/V^{3+}$  couple (Tables S4–S6†). These results also suggest that the catalytic effects of the surface functional groups on the anolyte and catholyte are different. Electrochemical impedance spectroscopy (EIS) data support our claims, as shown in Fig. S7.<sup>†</sup> For the redox reaction of  $V^{2+}/V^{3+}$ couple, P-CFs-400 exhibited the lowest charge transfer resistance in all the tested samples including CFs, O-CFs-400 and N-CFs-400. In contrast, P-CFs-400, O-CFs-400 and N-CFs-400 showed the similar charge transfer resistance on the redox reaction of the  $VO_2^{+}/VO_2^{+}$  couple. Additionally, the cycling stability of P-CFs-400 was tested by repetitive CV tests at a sweep rate of 10 mV s<sup>-1</sup> (Fig. S8<sup>†</sup>). During 200 cycles, the CV curves of P-CFs-400 were slightly changed for the redox reactions of both  $V^{2+}/V^{3+}$  and  $VO^{2+}/VO_{2}^{+}$  couples, indicating good cycling stability of P-CFs-400. The feasibility of using P-CFs-400 as the anolyte was further tested in full-cell VRFB systems (Fig. S8<sup>†</sup>), which were assembled with CFs as the counter electrode for the catholyte, and the electrochemical performance was compared with that of all the CFs systems. The galvanostatic chargedischarge profiles of P-CFs-400//CFs characterized in the operating voltage window of 0.8-1.6 V at a current density of 60 mA  ${\rm cm}^{-2}$  showed lower overpotentials of  ${\sim}50$  and  ${\sim}60$  mV in the charge and discharge profiles, respectively, compared with those of CFs//CFs (Fig. 3(c)). Therefore, under the same given conditions, P-CFs-400//CFs showed an increased capacity of  $\sim$ 29.8 A h L<sup>-1</sup> compared with that ( $\sim$ 26.8 A h L<sup>-1</sup>) of CFs//CFs, and the gap was more pronounced with an increase in the current density, as shown in Fig. S8(a).† In addition, the voltage efficiency of P-CFs-400//CFs was significantly higher than that of CFs//CFs (Fig. S8(b)<sup>†</sup>). As a result, the catalytic effects of the pyridonic-N groups led to improved energy efficiency of P-CFs-400//CFs by reducing the polarization for the redox reaction of the  $V^{2+}/V^{3+}$  couple (Fig. 3(d)). Although the energy efficiency of P-CFs-400//CFs is slightly higher than them of the previously reported values, the results shows the catalytic effects of P-CFs-400 in the anodic voltage region (Table S8<sup>+</sup>).<sup>46,47</sup>

### Conclusions

In summary, the surface of CF was dual-coated with oxygen and nitrogen-rich pyroproteins (P-CF) and the catalytic effects of the composites on the redox reaction of the  $V^{2+}/V^{3+}$  couple were compared with those of O-CFs and N-CFs. Whereas the oxygen groups on the surface of CFs have an insignificant catalytic effect on the  $V^{2+}/V^{3+}$  redox reaction, nitrogen groups led to a noticeable decrease of the  $\Delta E_p$  values in a nitrogen-concentration dependent manner, indicating superior catalytic effects of the nitrogen groups compared with oxygen on the

 $V^{2^+}/V^{3^+}$  redox couple. In contrast, P-CFs-400, with a lower nitrogen content than N-CFs-400, had the lowest  $\Delta E_p$  value of ~170.9 mV among the evaluated samples. This result suggests that the nitrogen and oxygen groups have a synergistic catalytic effect, especially the pyridonic groups, which are the main chemical structures in P-CFs-400. The *ex situ* XPS O 1s spectra showed a strong peak shift (~1.1 eV) for P-CFs-400, whereas that of O-CFs-400 exhibited a slight peak shift (~0.3 eV) despite the similarity of their oxygen configurations. These results support the proposed synergistic catalytic effects of oxygen and nitrogen in the pyridonic-*N* structure. Furthermore, the feasibility of the P-CF-based electrode was demonstrated in a full-cell VRFB system, where significantly improved energy efficiency was achieved compared with that of all CF-based electrodes.

### Conflicts of interest

There are no conflicts to declare.

### Acknowledgements

This work was financially supported by WCSL (World Class Smart Lab) research grant directed by Inha University.

### References

- 1 S. Chu and A. Majumar, Nature, 2012, 488, 294.
- 2 B. Dunn, H. Kamath and J.-M. Tarascon, *Science*, 2011, 334, 928.
- 3 D. Larcher and J.-M. Tarascon, Nat. Chem., 2015, 7, 19.
- 4 W. Wang, Q. Luo, B. Li, X. Wei, L. Li and Z. Yang, *Adv. Funct. Mater.*, 2013, **23**, 970.
- 5 K. Lin, Q. Chen, M. R. Gerhardt, L. Tong, S. B. Kim,
  L. Eisenach, A. W. Valle, D. Hardee, R. G. Gordon,
  M. J. Aziz and M. P. Marshak, *Science*, 2015, 349, 1529.
- 6 M. Park, J. Ryu, W. Wang and J. Cho, *Nat. Rev. Mater.*, 2016, 2, 16080.
- 7 D. R. Lide, *CRC Handbook of Chemistry and Physics*, CRC Press, Bora Raton, FL, USA, 2007.
- 8 F. F. Abdi, L. Han, A. H. M. Smets, M. Zeman, B. Dam and R. van de Krol, *Nat. Commun.*, 2013, 4, 2195.
- 9 R. M. Darling, A. Z. Weber, M. C. Tucker and M. L. Perry, *J. Electrochem. Soc.*, 2016, **163**, A5014.
- 10 J. Austing, C. N. Kirchner, L. Komsiyska and G. Wittstock, *J. Power Sources*, 2016, **306**, 692.
- 11 M. Skyllas-Kazacos, M. Rychcik, R. G. Robins, A. G. Fane and M. Green, *J. Electrochem. Soc.*, 1986, 133, 1057.
- 12 M. Skyllas-Kazacos and G. Grossmith, *J. Electrochem. Soc.*, 1987, **134**, 2950.
- 13 E. Sum and M. Skyllas-Kazacos, *J. Power Sources*, 1985, **15**, 179.
- 14 E. Sum, M. Rychcik and M. Skyllas-Kazacos, *J. Power Sources*, 1985, **16**, 85.
- 15 B. Li, M. Gu, Z. Nie, Y. Shao, Q. Luo, X. Wei, X. Li, J. Xiao, C. Wang, V. Sprenkle and W. Wang, *Nano Lett.*, 2013, 13, 1330.

- 16 B. Li, M. Gu, Z. Nie, X. Wei, C. Wang, V. Sprenkle and W. Wang, *Nano Lett.*, 2014, 14, 158.
- 17 M. Park, I.-Y. Jeon, J. Ryu, J.-B. Baek and J. Cho, *Adv. Energy Mater.*, 2015, 5, 1401550.
- 18 Y. Zhou, L. Liu, Y. Shen, L. Wu, L. Yu, F. Liang and J. Xi, *Chem. Commun.*, 2017, **53**, 7565.
- 19 M. Park, J. Ryu, Y. Kim and J. Cho, *Energy Environ. Sci.*, 2014, 7, 3727.
- 20 J. Kim, H. Lim, J.-Y. Jyoung, E.-S. Lee, J. S. Yi and D. Lee, *Carbon*, 2017, **111**, 592.
- 21 A. M. Pezeshki, J. T. Clement, G. M. Veith, T. A. Zawodzinski and M. M. Mench, *J. Power Sources*, 2015, **294**, 333.
- 22 J. H. Park, J. J. Park, O. O. Park and J. H. Yang, *ChemSusChem*, 2016, 9, 1.
- 23 K. J. Kim, H. S. Lee, J. Kim, M.-S. Park, J. H. Kim, Y.-J. Kim and M. Skyllas-Kazacos, *ChemSusChem*, 2016, 9, 1329.
- 24 T. Liu, X. Li, H. Nie, C. Xu and H. Zhang, J. Power Sources, 2015, 286, 73.
- 25 D. Dixon, D. J. Babu, J. Langner, M. Bruns, L. Pfaffmann, A. Bhaskar, J. J. Schneider, F. Scheiba and H. Ehrenberg, *J. Power Sources*, 2016, 332, 240.
- 26 J. J. Park, J. H. Park, O. O. Park and J. H. Yang, *Carbon*, 2016, 110, 17.
- 27 S. Jeong, S. Kim and Y. Kwon, *Electrochim. Acta*, 2013, **114**, 439.
- 28 K. J. Kim, M.-S. Park, J.-H. Kim, U. Hwang, N. J. Lee, G. Jeong and Y.-J. Kim, *Chem. Commun.*, 2012, **48**, 5455.
- 29 D. M. Kabtamu, J.-Y. Chen, Y.-C. Chang and C.-H. Wang, *J. Mater. Chem. A*, 2016, **4**, 11472.
- 30 M. Jing, X. Zhang, X. Fan, L. Zhao, J. Liu and C. Yan, *Electrochim. Acta*, 2016, **215**, 57.
- 31 H. Zhou, Y. Shen, J. Xi, X. Qiu and L. Chen, *ACS Appl. Mater. Interfaces*, 2016, **8**, 15369.

- 32 S. Wang, X. Zhao, T. Cochell and A. Manthiram, J. Phys. Chem. Lett., 2012, 3, 2164.
- 33 S. Park and H. Kim, J. Mater. Chem. A, 2015, 3, 12276.
- 34 L. Estevez, D. Reed, Z. Nie, A. M. Schwarz, M. I. Nandasiri,
  J. P. Kizewski, W. Wang, E. Thomsen, J. Liu, J.-G. Zhang,
  V. Sprenkle and B. Li, *ChemSusChem*, 2016, 9, 1455.
- 35 K. J. Kim, S.-W. Lee, T. Yim, J.-G. Kim, J. W. Choi, J. H. Kim, M.-S. Park and Y.-J. Kim, *Sci. Rep.*, 2014, 4, 6906.
- 36 C. Flox, J. Rubio-García, M. Skoumal, T. Andreu and J. R. Morante, *Carbon*, 2013, **60**, 280.
- 37 L. Wu, Y. Shen, L. Yu, J. Xi and X. Qiu, *Nano Energy*, 2016, 28, 19.
- 38 M. Park, Y.-J. Jung, J. Kim, H. I. Lee and J. Cho, *Nano Lett.*, 2013, **13**, 4833.
- 39 D. J. Suárez, Z. González, C. Banco, M. Granda, R. Menéndez and R. Santamaría, *ChemSusChem*, 2014, 7, 914.
- 40 T.-M. Tseng, R.-H. Huang, C.-Y. Huang, C.-C. Liu, K.-L. Hsueh and F.-S. Shieu, *J. Electrochem. Soc.*, 2014, **161**, A1132.
- 41 Y. S. Yun, K.-Y. Park, B. Lee, S. Y. Cho, Y.-U. Park, S. J. Hong,
  B. H. Kim, H. Gwon, H. Kim, S. Lee, Y. W. Park, H.-J. Jin and
  K. Kang, *Adv. Mater.*, 2015, 27, 6914.
- 42 J. H. Park, J. J. Park, O. O. Park, C.-S. Jin and J. H. Yang, *J. Power Sources*, 2016, **137**, 137.
- 43 K. N. Kudin, B. Ozbas, H. C. Schniepp, R. K. Prud'homme, I. A. Aksay and R. Car, *Nano Lett.*, 2008, **8**, 36.
- 44 M. Lee, J. Hong, D.-H. Seo, D. H. Nam, K. T. Nam, K. Kang and C. B. Park, *Angew. Chem., Int. Ed.*, 2013, **52**, 8322.
- 45 Y. S. Yun, D.-H. Kim, S. J. Hong, M. H. Park, Y. W. Park,
  B. H. Kim, H.-J. Jin and K. Kang, *Nanoscale*, 2015, 7, 15051.
- 46 B. Jiang, L. Wu, L. Yu, X. Qiu and J. Xi, *J. Membr. Sci.*, 2016, 510, 18.
- 47 L. Wu, J. Wang, Y. Shen, L. Liu and J. Xi, *Phys. Chem. Chem. Phys.*, 2017, **19**, 14708.